Insertion of Trivalent Lanthanides into Uranyl Vanadate Layers and Frameworks

Yaxing Wang,†,‡ Xuemiao Yin,†,‡ Yanyan Zhao,†,‡ Yang Gao,†,‡ Lanhua Chen,†,‡ Zhiyong Liu,†,‡ Daopeng Sheng,†,‡ Juan Diwu,†,‡ Zhifang Chai,†,‡ Thomas E. Albrecht-Schmitt,§ and Shuao Wang*,†,‡

 † School for Radiological and Interdisciplinary Sciences (RAD-X) and ‡ Collaborative Innovation Center of Radiation Medic[ine](#page-5-0) of Jiangsu Higher Education Institutions, Soochow University, 199 Renai Road, Suzhou Industrial Park, Suzhou 215123, People's Republic of China

§ Department of Chemistry and Biochemistry, Florida State University, Tallahassee, Florida 32306, United States

S Supporting Information

[AB](#page-5-0)STRACT: [Two new ur](#page-5-0)anyl vanadates have been prepared from hydrothermal reactions and structurally characterized by single-crystal X-ray diffraction. The structure of $(H_3O)UO_2VO_4$ (UVO-1) consists of anionic layers containing UO_2^{2+} pentagonal bipyramids coordinated by edge-sharing $VO₅$ square pyramids, with the charge balanced by interlaminar H_3O^+ cations. Vanadium in $(UO_2)_3(VO_4)_2(H_2O)_3$ (UVO-2) exists as monomeric VO₄ tetrahedra coordinating to $\mathrm{UO_2}^{2+}$ pentagonal bipyramids, forming

a 3D uranyl(VI) vanadate framework. Similar reactions with the addition of $Ln(NO₃)₃$ (Ln = Nd, Eu) afford the three heterobimetallic lanthanide uranyl vanadate frameworks $Nd(UO_2)_3(VO_4)_3(H_2O)_{11}$ (NdUVO-1), Eu(UO₂)₃(VO₄)₃(H₂O)₁₀ (EuUVO-1), and $Eu(UO_2)_{12}(VO_4)_{10}(H_2O)_{24}$ (EuUVO-2). In NdUVO-1 and EuUVO-1, Ln³⁺ cations are inserted into the interlayer space of UVO-1 substituting for H_3O^+ and further bridging adjacent layers into 3D frameworks. Similarly, EuUVO-2 adopts the same sheet topology as UVO-2, with $Eu³⁺$ ions replacing some of the interlayer uranyl ions in UVO-2. Our work has demonstrated that uranyl vanadate extended structures are excellent hosts for further incorporation of trivalent lanthanide/ actinide cations and has provided a new way to create new heterobimetallic 4f−5f and 5f−5f compounds.

1. INTRODUCTION

There is renewed interest in nuclear power in many technologically advanced nations. Deep geological disposal is a preferred option for the management of nuclear waste in many countries, especially for the vast amounts of used nuclear fuel and other high-level radioactive waste.^{1a−c} However, release of high toxicity and long-half-life radionuclides during long-term storage is a primary concern. Fu[rthe](#page-6-0)rmore, the migration of hazardous radionuclides creates a significant threat to the environment and public safety.^{1c,d} Burns et al. proposed that uranyl compounds formed during corrosion and oxidative alteration of used nuclear fuel can [furt](#page-6-0)her act as a host to incorporate transuranium elements by charge-balance substitution, which may decrease the mobility of these radionuclides.² A new proposed strategy for immobilization of soluble actinide elements utilizing coprecipitation methods at low tem[pe](#page-6-0)rature has been accomplished in recent work; actinyl ions (i.e., NpO_2^+ and UO_2^2) have been successfully incorporated in natural minerals. 3 The sequestration of two key cocontaminants in nuclear waste, ^{237}Np and ^{129}I , has been accomplished by cocrystallizatio[n](#page-6-0) within uranyl(VI) com $pounds_i⁴$ but little is known about the incorporation of trivalent minor actinides (americium and curium) within uranyl compo[un](#page-6-0)ds. The separation and sequestration or fissioning of americium is a key component of closed, advanced nuclear fuel cycles. Trivalent lanthanides (primarily Nd^{3+} and Eu^{3+}) have been used as analogues of trivalent transplutonium elements (e.g., Am^{3+} and Cm^{3+}) owing to their nearly identical ionic radii. Therefore, understanding the cocrystallization of trivalent lanthanides with uranyl in heterobimetallic 4f−5f compounds is important for designing new advanced waste forms and for predicting the fate of actinides in the environment. There are only a handful of reports in the literature concerning this family of compounds. Volkringer et al. reported several heterobimetallic uranyl/lanthanide mellitates, 5 which can be used as thermal degradation precursors of mixed 4f−5f or 5f−5f oxide solid solutions. Knope et al. and [L](#page-6-0)iu et al. synthesized heterobimetallic uranyl/lanthanide carboxyphosphonates and germanates, respectively, with interesting photoluminescence sensitization phenomena.⁶ A series of ordered and disordered 4f−5f and 5f−5f heterobimetallic diphosphonates were prepar[ed](#page-6-0) and characterized by Diwu et al.⁷

Recently, the formation of highly insoluble vanadate minerals has been shown to be effective in the rem[e](#page-6-0)diation of uraniumcontaminated groundwater and appears to be superior to phosphate precipitation in many regards.⁸ The origin of these improvements lies in the enhanced coordinating abilities of vanadates to uranyl (VI) , which is typic[al](#page-6-0)ly found within the UO_2^{2+} moiety. These natural phases control the solubility of

Received: May 20, 2015 Published: August 20, 2015

U(VI) even when phosphate is present and are solubilitylimiting products. Owing to the formation of these minerals over a wide range of conditions, a large family of $uranyl(VI)$ vanadate minerals have been identified that include carnotite, curienite, francevillite, tyuyamunite, and metatyuyamunite.⁹ Combining variable oxidation states of vanadium and the structural diversity of vanadates, crystalline actinide vanadate[s](#page-6-0) could be potential candidates for effectively immobilizing multiple actinide radioisotopes under different environmental conditions,¹⁰ especially for emergent treatment for water systems contaminated with small amounts of uranium and minor acti[nid](#page-6-0)es.⁸ A series of synthetic uranyl vanadates have also been reported and extensively characterized, with the majority adopti[ng](#page-6-0) 2D anionic layered structures with alkalimetal cations residing in the interlayer space.¹¹ These aforementioned compounds were synthesized by high-temperature, solid-state reactions. In addition, several [org](#page-6-0)anically templated uranyl vanadates have been prepared by traditional hydrothermal methods.¹²

Herein, we report two uranyl(VI) vanadate compounds with different structural top[olo](#page-6-0)gies that can be used as crystalline hosts for further incorporation of trivalent lanthanides via substitution of components present in the interlayer space, confirmed by X-ray crystallography, UV−vis absorption spectroscopy, energy-dispersive X-ray spectroscopy, and laserablation inductively coupled plasma mass spectrometry. Our work is not only helpful in creating new heterobimetallic 4f−5f or 5f−5f compounds but also provides structural information for a further understanding of the incorporation mechanism of trivalent minor actinides into uranyl vanadate minerals in the environment.

2. EXPERIMENTAL SECTION

Caution! All uranium compounds used in these studies contained depleted uranium; standard precautions were performed for handling radioactive materials, and all studies were conducted in a laboratory dedicated to studies on actinide elements.

2.1. Reagents. V_2O_3 , $Eu(NO_3)_3.6H_2O$, $Nd(NO_3)_3.6H_2O$, and $UO₂(CH₃COO)₂$ ²H₂O were used as received.

2.2. Synthesis of UVO-1 and UVO-2. A 0.1281 g portion of $UO_2(CH_3COO)_2.2H_2O$, 0.0236 g of V_2O_3 (U:V molar ratio 1:1), and 2 mL of ultrapure water were loaded into a 10 mL autoclave. The

autoclave was then sealed and heated to 200 °C for 3 days under autogenous pressure and cooled to room temperature at a rate of 3.5 $\rm{^{\circ}C/h}$. The product consists of yellow prismatic crystals of $\rm(H_{3}O)$ - UO_2VO_4 (abbreviated UVO-1) as the minor phase and yellow acicular crystals of $(UO_2)_3(VO_4)_2(H_2O)_3$ (abbreviated UVO-2) as the major phase, as well as a trace amount of unreacted black V_2O_3 .

2.3. Synthesis of NdUVO-1, EuUVO-1, and EuUVO-2. A 0.1281 g portion of $(UO_2)(CH_3COO)_2$, 0.0236 g of V_2O_3 , 0.1338 g of $Nd(NO₃)₃·6H₂O$ (or 0.1355 g of Eu(NO₃)₃·6H₂O, Ln:U:V molar ratio 1:1:1), and 2 mL of ultrapure water were loaded into 10 mL autoclaves. The autoclaves were then sealed and heated to 200 °C for 3 days under autogenous pressure and cooled to room temperature at a rate of 3.5 °C/h. Isotypic yellow platelike crystals of Nd- $(UO₂)₃(VO₄)₃(H₂O)₁₁$ (abbreviated NdUVO-1) and Eu- $(UO₂)₃(VO₄)₃(H₂O)₁₀$ (abbreviated EuUVO-1) were isolated, respectively. Eu $(UO_2)_3(VO_4)_3(H_2O)_{12}$ (abbreviated EuUVO-2) was obtained from a reaction extended over 4 days.

2.4. Crystallographic Studies. Crystals were mounted on Cryoloops with Paratone and optically aligned on a Bruker D8- Venture single-crystal X-ray diffractometer equipped with a digital camera. The diffraction data were collected using a Turbo X-ray Source (Mo K α radiation, $\lambda = 0.71073$ Å) adopting the direct-drive rotating anode technique and a CMOS detector at room temperature. The structures were solved by direct methods and refined on F^2 by full-matrix least-squares methods using SHELXTL. 13 For UVO-1, all of the non-hydrogen atoms were refined anisotropically and the hydrogen atoms of water molecules were found [fr](#page-6-0)om the Fourier maps. For UVO-2, only atoms (including U, V, and O) present in the 2D uranyl vanadate host layers (see Results and Discussion in the main text) could be refined anisotropically, while the uranium atoms located at the interlayer spaces were partially disordered and could not be refined anisotropically. The disorde[r treatment can be found](#page-2-0) in the CIF files in the Supporting Information. For NdUVO-1 and EuUVO-1, only atoms present in the 2D uranyl vanadate host layers could be refined anisotropically, similarly to the case for UVO-2. The [incorpora](http://pubs.acs.org/doi/suppl/10.1021/acs.inorgchem.5b01141/suppl_file/ic5b01141_si_002.cif)ted lanthanide atoms are partially disordered. For EuUVO-2, only uranium atoms could be refined anisotropically, including those in the interlayer spaces. Again, the incorporating lanthanide atoms are partially disordered. The presenting A-level alerts in the checkcif file for EuUVO-2 are not reasonable, since the maximum shift we observed during refinement was 0.000 Å; however, this could originate from difficulties in refining the coordinating water molecules around disordered $Eu³⁺$ cations. Attempts to increase the refinement cycles or adjust the site occupancy correspondingly did not solve this issue. Crystallographic data for UVO-1, UVO-2, NdUVO-1, EuUVO-1, and EuUVO-2 are given in Table 1.

2.5. Powder X-ray Diffraction. Powder patterns were collected from 5 to 50°, with a step of 0.02° using a Bruker D8 advance X-ray diffractometer with Cu Ka radiation ($\lambda = 1.54056$ Å) equipped with a Lynxeye one-dimensional detector.

2.6. UV−Vis Absorption and Photoluminescence Spectroscopy. UV−vis absorption spectra were recorded from single crystals of five phases using a Craic Technologies microspectrophotometer. Crystals were placed on quartz slides, and the data were collected after optimization of the microspectrophotometer. Photoluminescence spectroscopy data were recorded from single crystals of lanthanideincorporated phases using a Craic Technologies microspectrophotometer. Crystals were placed on quartz slides, and the data were collected after optimization of the microspectrophotometer.

2.7. Laser-Ablation Inductively Coupled Plasma Mass Spectrometry. LA-ICP-MS measurements on single crystals from lanthanide-incorporated phases were conducted using a ThermoFisher Element2 ICP-MS instrument coupled to a UP213 Nd:YAG laser ablation system (New Wave Research).

2.8. Energy-Dispersive X-ray Spectroscopy Analysis. Scanning electron microscopy/energy-dispersive spectroscopy (SEM/ EDS) images and data were collected using a FEI Quanta 200FEG instrument. The energy of the electron beam was 30 kV, and the spectrum acquisition time was 100 s.

2.9. IR Spectrum. The infrared spectrum of UVO-1 was recorded using a single crystal with a Bruker Hyperion instrument in the range of 600–4000 cm^{-1} . .

3. RESULTS AND DISCUSSION

3.1. Synthesis. The structure of vanadate oxoanion compounds in hydrothermal systems is greatly controlled by the reaction conditions, such as pH and temperature. Different polymerized vanadate oxoanion species can be obtained by adjusting the solution acidity.¹⁰ In the present work, the reaction of V_2O_3 with uranyl acetate by traditional hydrothermal methods (typically 20[0](#page-6-0) °C and 72 h) results in two distinct uranyl vanadates, the minor phase being yellow prismatic crystals of UVO-1 and the major phase yellow acicular crystals of UVO-2. Trace amounts of unreacted black powder of V_2O_3 are present in the product as well (a microscopic image is shown in Figure S1 in the Supporting Information). Obviously, most of the trivalent vanadium source is oxidized to pentavalent vanad[ates, possib](http://pubs.acs.org/doi/suppl/10.1021/acs.inorgchem.5b01141/suppl_file/ic5b01141_si_001.pdf)ly by O_2 in the air during the reaction process. A prolonged reaction time (120 h) has no influence on the product. The heterobimetallic 4f−5f compounds NdUVO-1 and EuUVO-1 can be obtained under the aforementioned reaction conditions with the addition of 1 equivalent of lanthanide nitrate. Two isotypic compounds can be collected as the pure phases, as demonstrated by the powder X-ray diffraction data (Figure S2 in the Supporting Information). EuUVO-2 can be isolated as a pure phase by extending the reaction time [to 96 h. Ho](http://pubs.acs.org/doi/suppl/10.1021/acs.inorgchem.5b01141/suppl_file/ic5b01141_si_001.pdf)wever, only NdUVO-1 was obtained from prolonged reaction time (PXRD data are shown in Figure S3 in the Supporting Information). Therefore, it was speculated that the subtle difference in ionic radii between Nd^{3+} [and](http://pubs.acs.org/doi/suppl/10.1021/acs.inorgchem.5b01141/suppl_file/ic5b01141_si_001.pdf) Eu^{3+} greatly affected the reaction process and products.

3.2. Crystal Structures. Single-crystal X-ray diffraction studies revealed that UVO-1 is isotypic with $(NH_4)_2[(UO_2)_2V_2O_8]^{12b}$ and crystallizes in the centrosymmetric monoclinic space group $P2₁/c$. The overall structure is based on anionic laye[rs;](#page-6-0) the crystallographically ordered H_3O^+ distributes in the interlayer space to balance the negative charge of the layers, as shown in Figure 1. The presence of hydronium cation is further confirmed by two weak features at 1680 and 1720 cm[−]¹ as well as a broad feature between 3000 and 3500

Figure 1. View of the crystal structure of UVO-1 along the *b* axis: H_3O^+ , light blue; UO₇. yellow; dimers of VO₅, purple.

cm[−]¹ in the FTIR micro spectrum measured from a single crystal of $UVO-1$ (Figure S4 in the Supporting Information).¹⁴ The layers of UVO-1 are parallel to the $[bc]$ plane and adopt the carnotite topology, 15 which contain dimers of edge-shari[ng](#page-6-0) $UO₇$ pentagonal b[ipyramids](http://pubs.acs.org/doi/suppl/10.1021/acs.inorgchem.5b01141/suppl_file/ic5b01141_si_001.pdf) and dimers of edge-sharing $VO₅$ square pyramids. Wit[hin](#page-6-0) the $VO₅$ dimers, the vertices of two adjacent square pyramids point in opposite directions. There is only one crystallographically independent uranium site and one vanadium site in the asymmetric unit. The U−O equatorial bond lengths range from 2.297(3) to 2.376(4) Å, whereas the axial U \equiv O bonds are 1.806(4) and 1.810(4) Å, respectively. The V−O bond lengths within the square plane range from 1.792(4) to 1.930(4) Å, and the short vertex bond $(1.612(4))$ Å) can be assigned as a vanadyl unit, in comparison to typically observed vanadyl bonds that occur in the range 1.56−1.62 Å.¹⁶ All five equatorial oxygen atoms of uranyl are μ_3 -O atoms. The bond valence sums (BVS) calculated for the three oxyg[en](#page-6-0) atoms in the asymmetric unit are all close to 2 (O1, 1.92; O3, 1.89; O4, 1.70), indicating their full deprotonation.

In contrast, UVO-2 crystallizes in the orthorhombic system in the space group Pnma. Its structure can be best described as a porous 3D framework constructed from uranophane-type layers containing $UO₇$ pentagonal bipyramids and monomeric VO4 distorted tetrahedra that are further bridged through additional interlaminar $UO₇$ polyhedra, as depicted in Figure 2. The asymmetric unit of UVO-2 consists of four independent U sites and two V sites. The axial and equatorial uranium oxygen bonds range from 1.766(11) to 1.786(12) Å and from 2.316(8)

Figure 2. View of the crystal structure of UVO-2 down the c axis: $UO₇$, yellow; $VO₄$, purple. The linkage $UO₇$ in the interlayer along b axis is distorted with larger crystallographic thermal parameters.

Figure 3. (a) View of the crystal structure of LnUVO-1 along the b axis: UO_7 , yellow; LnO_n, green; dimers of VO₅, purple (vanadates without lanthanide coordination) or dark blue (vanadates with lanthanide coordination). (b) Ball-and-stick model for coordinated V=O and terminal V=O in EuUVO-1.

to 2.60(3) Å, respectively, while V–O distances within the VO_4 unit range from $1.651(12)$ to $1.732(8)$ Å. The uranophane-type layers are built through $UO₇$ edge-sharing chains (U1 and U2) connected via $VO₄$ tetrahedra in a corner-sharing mode with $UO₇$ units in their equatorial planes. The other two uranyl sites (U3 and U4) are coordinated to these $VO₄$ tetrahedra in the direction perpendicular to the layers (along the b axis), thus bridging adjacent layers to afford an open-framework structure with two types of 1D channels extended along the c axis. The sizes of the channels are 6.206×5.275 Å and 4.461×3.244 Å, respectively. Notably, the bridging uranium sites are partially disordered, strongly indicating weaker interlayer interactions between vanadates and bridging uranyl groups in comparison to the corresponding intralayer coordination. Along with the relatively larger crystallographic thermal parameter (U_{eq}) of hydronium oxygen in comparison to others in UVO-1, the crystallographic data therefore indicated that both uranyl vanadate compounds could be used as excellent host lattices to incorporate other metal cations in the interlayer space.

The addition of 1 equivalent of $Nd(NO₃)₃·6H₂O$ or $Eu(NO₃)₃·6H₂O$ to the reaction mixtures (Ln:U:V molar ratio 1:1:1) without a change in any other reaction parameters results in the formation of NdUVO-1 and EuUVO-1. The difference in numbers of coordinated waters binding to the Ln^{3+} center is partially due to the subtle change in ionic radii. The crystal structure of $LnUVO-1$ ($Ln = Nd$, Eu) consists of a 4f−5f heterobimetallic 3D framework built from uranyl vanadate anionic layers linked by interlaminar hydrated $Ln³⁺$ cations, as shown in Figure 3. LnUVO-1 crystallizes in the monoclinic space group $P2_1/n$. The asymmetric unit of EuUVO-1 contains three independent U sites, three V sites, and one Ln site; the uranyl cation also adopts a $UO₇$ pentagonal-bipyramidal coordination geometry, in which five equatorial oxygen atoms are supplied by a $VO₅$ square pyramid.

The vanadyl units with relatively short $V=O$ bonds provide oxygen atoms to bind to distorted lanthanide cations, constructing a 4f−5f heterobimetallic three-dimensional framework. The U≡O and U−O bond lengths are in the ranges of 1.795(15)−1.850(15) and 2.300(15)−2.384(15) Å, respectively. The significant change in bond length of $V=O$ could be clearly observed in vanadyl units. As shown Figure 3b, bond lengths in V= O coordinating to Eu³⁺ centers (V(1)−O(18) 1.624(15) Å, V(2)−O(16) 1.601(15) Å) are greatly elongated in comparison with the terminal V=O length $(V(3)-O(17))$ $1.578(17)$ Å), confirming the Lewis basic nature of vanadyl oxygen being similar to that of the bridging uranyl oxygen, formally called cation−cation interactions (CCIs).¹⁷ To the best of our knowledge, this is the first report of the formation of V=O−Ln bonds, further suggesting the potential a[pp](#page-6-0)lications of remediating heavy-metal pollutants using materials containing coordination-available vanadyl oxygen donors.¹⁸ The V=O bond lengths in vanadyl units are summarized in Table 2.

Interestingly, the topology, composition, and a[rra](#page-6-0)ngement of uranyl vanadate layers in LnUVO-1 closely resemble those of UVO-1, with subtle differences induced by lanthanide insertion (Figure 5a,b). Actually, LnUVO-1 crystallizes in the space group $P2_1/n$, which is a transformation from the space group of [UVO-1,](#page-4-0) $P2₁/c$. In view of the same reaction conditions and the structural similarities between UVO-1 and LnUVO-1, interlayer cations are likely to compete with each other during the formation reactions of LnUVO-1: i.e., the hydronium cations in UVO-1 are substituted by Ln^{3+} cations in the structures. As a result of lanthanide incorporation, the number of independent sites of uranium and vanadium within the layers increases from one to three in LnUVO-1, with two-thirds of the $VO₅$ square pyramids providing vanadyl oxygen to coordinate lanthanide ions. Meanwhile, the presence of lanthanide ions in the interlayer expands the distance between adjacent layers from

Inorganic Chemistry Article 2018 **Article 2018 Article 2018 Article 2018 Article 2019 Articl**

6.992 Å in UVO-1 to 7.935 and 7.867 Å in NdUVO-1 and EuUVO-1, respectively. The difference between the last two agrees well with the phenomenon of lanthanide contraction.^{12b} The coordination spheres of the incorporating lanthanides are completed by eight additional coordinated water molec[ules](#page-6-0) with slight disorder.

For UVO-2, which possesses a framework structure, the partially disordered interlayer uranyl ions can also be substituted with prolonged reaction times in the presence of $Eu(NO₃)₃·6H₂O$. **EuUVO-2** crystallizes in a triclinic crystal system in the P1 space group and is built from uranophane-type vanadate layers bridged by additional uranyl and $Eu³⁺$ polyhedra, forming a dense three-dimensional framework, as shown in Figure 4. The asymmetric unit of EuUVO-2 consists

Figure 4. View of crystal structure of **EuUVO-2** down the c axis: UO_7 , yellow; EuO_w green; $VO₄$, purple.

of 12 independent U sites, 10 V sites, and 2 Eu sites. The framework can be divided into two components: a $\mathrm{UO_2VO_4}^$ layer and $\mathrm{Eu}^{3+}/\mathrm{UO_2}^{2+}$ ions as the linkages. The $\mathrm{UO_2VO_4}^-$ layer is constructed from edge-sharing $UO₇$ pentagonal bipyramids, with three equatorial oxygen atoms provided by the monomeric $\rm VO_4$ tetrahedron. The linkage $\rm{UO_2}^{2+}$ ions adopt the coordination geometry of a $UO₇$ pentagonal bipyramid with O \equiv U \equiv O moieties parallel with the UO₂VO₄⁻ layer; three equatorial oxygen atoms are provided by $VO₄$ tetrahedra, and the other two are coordinated water. The other linkage Eu^{3+} ions are also bonded by two $VO₄$ tetrahedra, while $Eu³⁺$ ions exhibit a distorted-polyhedral coordination geometry with weakly coordinated water. Therefore, a dense three-dimensional framework with low symmetry is constructed. As found in the previous example, the topology of EuUVO-2 is identical with that of UVO-2 (Figure 5c,d)) with partial interlayer uranyl ions in UVO-2 substituted by Eu^{3+} cations, forming a denser 3D framework in comparison to that of UVO-2. The distance between adjacent layers in EuUVO-2 is slightly expanded, owing to the larger ionic radius of Eu³⁺ in comparison to that of uranyl ions. The uranium atoms residing in the interlayer spaces are fully ordered with the $UO₇$ pentagonal-bipyramidal geometry, while the incorporating $Eu³⁺$ cations are partially disordered, strongly indicating partial substitution of disordered interlaminar uranium in UVO-2 with Eu^{3+} in EuUVO-2.

3.3. SEM/EDS and LA-ICP-MS Analysis. For further confirmation of the incorporation of lanthanide cations in the uranyl vanadate compounds, SEM/EDS measurements were performed on single crystals. SEM images of four crystals are

Figure 5. Layer topologies of UVO-1 (a), NdUVO-1 (b), UVO-2 (c), and EuUVO-2 (d): UO_7 , yellow; dimers of VO_5 in UVO-1 and NdUVO-1, purple. Monomeric $VO₄$ units in UVO-2 and EuUVO-2 are also represented in purple. The dark blue in NdUVO-1 represents vanadates with lanthanide coordination.

shown in Figure 6. Energy-dispersive X-ray spectroscopy of four compounds (Table 3) clearly shows that the atomic ratio is

Figure 6. SEM images of UVO-1, UVO-2, NdUVO-1, and EuUVO-2.

consistent with the crystallographic results, indicating that lanthanide cations are successfully incorporated into uranyl vanadates. Moreover, crystals of EuUVO-1, EuUVO-1, and EuUVO-2 were selected for laser-ablation inductively coupled plasma mass spectrometry to investigate the depth distribution of lanthanides in the crystals of the heterobimetallic compounds. Figure S5 in the Supporting Information demonstrates a typical time-resolved spectrum (i.e., ion signal measured in [counts per](http://pubs.acs.org/doi/suppl/10.1021/acs.inorgchem.5b01141/suppl_file/ic5b01141_si_001.pdf) second (cps) vs time (s)). The background signal measurements show that background count rates are in range of 10−100 cps, while the neodymium signal exhibits enhanced counts when laser ablation was performed on the crystal. The neodymium signal is concomitant with the signal for uranium, indicating that Ln ions are incorporated into the entire crystals of uranyl vanadates instead of being adsorbed on the surface.¹⁹

Table 3. Energy-Dispersive X-ray Spectroscopy Results for Four Selected Compounds

		element/atom %				
	formula	Ln	U		O	theor mole ratio
$UVO-1$	$(H_3O)UO_2VO_4$		9.43	8.25	82.32	$U:V = 1:1$
$UVO-2$	$(UO_2)_{3}(VO_4)_{2}(H_2O)_{3}$		10.53	9.61	79.86	$U:V = 1.5:1$
NdUVO-1	$Nd(UO_2)_{3}(VO_4)_{3}(H_2O)_{11}$	3.88	10.01	10.97	75.13	$Nd:U:V = 1:3:3$
EuUVO-2	$Eu_2(UO_2)_{12}(VO_4)_{10}(H_2O)_{24}$	2.72	11.45	10.89	74.94	$Eu:U:V = 1:6:5$

3.4. Spectroscopic Properties. Solid-state UV−vis−NIR absorption spectrum of all compounds is shown in Figure 7.

The typical charge transfer absorption peaks of equatorial U−O and the axial U \equiv O are not observed in NdUVO-1, EuUVO-1, and UVO-1 due to large charge transfer features from $VO₅$ units dominating the UV/visible regions, 20 as shown in Figure 7a. Figure 7b provides clear evidence that Eu^{3+} ions are incorporated into uranyl vanadates; [ar](#page-6-0)rows indicate the characteristic f–f transitions for Eu^{3+ 20a} The two peaks from 200 to 500 nm are the vibronic progressions from uranyl and Eu3+. ²⁰ From 200 to 400 nm, there is [a s](#page-6-0)uperposition of V−O charge transfer and equatorial U−O charge transfer.²¹ The featu[res](#page-6-0) from 400 to 500 nm are the uranyl axial CT bands that are vibronically coupled. Unlike the case for $(UO₂)₂(PPA)$ $(HPPA)_{2}Sm(H_{2}O) \cdot 2H_{2}O$ and $K_{4}[(UO_{2})Eu_{2}(Ge_{2}O_{7})_{2}]^{11}$ uranyl sensitization for the lanthanide photoluminescence is not observed, likely due to the fluorescence quenching ori[gin](#page-6-0)ated

from crystallographic disorders of lanthanide cations and nonradiative energy loss through the bound water molecules.

4. CONCLUSION

In conclusion, two new uranyl vanadates have been synthesized, starting with V_2O_3 , and trivalent lanthanide cations have been successfully inserted into their layers and frameworks. This intercalation is achieved by combining the stability of the uranyl vanadate layers and the binding of the lanthanide cations by the vanadyl oxo atoms. The structures of LnUVO-1, featuring the first V $=$ O−Ln bond reported in a solid-state structure, further suggest the potential applications of remediating heavy-metal pollutants using materials containing coordination-available vanadyl oxygen donors. What we observed in this system are real crystal lattice sites possessed by a chemically robust uranyl vanadate extended structure for effective immobilization of trivalent lanthanides and actinides, which may lead to extra stabilization of minor actinides provided by lattice energy. Our work not only is helpful in creating new heterobimetallic 4f−5f or 5f−5f compounds but also provides structural information for further simultaneous immobilization of trivalent minor actinides with uranyl compounds using vanadate sources as well as further incorporation of trivalent minor actinides into uranyl vanadate minerals in the environment.

■ ASSOCIATED CONTENT

8 Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.inorgchem.5b01141.

[Powder X-ray di](http://pubs.acs.org)ffraction patterns, in[frared spectrum, and](http://pubs.acs.org/doi/abs/10.1021/acs.inorgchem.5b01141) [laser-abla](http://pubs.acs.org/doi/abs/10.1021/acs.inorgchem.5b01141)tion inductively coupled plasma spectra (PDF) X-ray crystallographic files for UVO-1, UVO-2, NdUVO-1, EuUVO-1, and EuUVO-2 (CIF)

■ AUTHOR INFORMATION

Corresponding Author

*E-mail for S.W.: shuaowang@suda.edu.cn.

Notes

The authors decla[re no competing](mailto:shuaowang@suda.edu.cn) financial interest.

■ ACKNOWLEDGMENTS

This work was supported by the National Natural Foundation of China (91326112, 21422704, 21471107), the Science Foundation of Jiangsu Province (BK20140007), a Project Funded by the Priority Academic Program Development of Jiangsu Higher Education Institutions (PAPD), and "Young Thousand Talented Program" in China. Support for TEA-S was provided by the Chemical Sciences, Geosciences, and Biosciences Division, Office of Basic Energy Sciences, Office of Science, Heavy Elements Chemistry Program, U.S. Department of Energy, under Grant DE-FG02-13ER16414.

(1) (a) Grambow, B. Elements 2006 2, 357. (b) Bruno, J.; Ewing, R. , C. Elements 2006 2, 343. (c) Baker, R. J. Coord. Chem. Rev. 2014 , 266- , 267, 123. (c) Ewing, R. C. Nat. Mater. 2015 , 14, 252. (d) Maher, K.; Bargar, J. R.; Brown, G. E., Jr. Inorg. Chem. 2013 , 52, 3510.

(2) Burns, P. C.; Ewing, R. C.; Miller, M. L. J. Nucl. Mater. 1997 , 245 , 1.

(3) Balboni, E.; Morrison, J. M.; Wang, Z.; Engelhard, M. H.; Burns, P. C. Geochim. Cosmochim. Acta 2015, 151, 133.

(4) (a) Meredith, N. A.; Polinski, M. J.; Lin, J.; Simonetti, A.; Albrecht-Schmitt, T. E. Inorg. Chem. 2012 , 51, 10480. (b) Wu, S.; Chen, F.; Simonetti, A.; Albrecht-Schmitt, T. E. Environ. Sci. Technol. 2010 , 44, 3192.

(5) (a) Mihalcea, I.; Volkringer, C.; Henry, N.; Loiseau, T. Inorg. Chem. 2012 , 51, 9610. (b) Volkringer, C.; Henry, N.; Grandjean, S.; Loiseau, T. J. Am. Chem. Soc. 2012 , 134, 1275.

(6) (a) Knope, K. E.; de Lill, D. T.; Rowland, C. E.; Cantos, P. M.; de Bettencourt-Dias, A.; Cahill, C. L. *Inorg. Chem.* **2012**, 51, 201. (b) Liu, S.-P.; Chen, M.-L.; Chang, B.-C.; Lii, K.-H. Inorg. Chem. 2013, 52 , 3990.

(7) Diwu, J.; Wang, S.; Good, J. J.; DiStefano, V. H.; Albrecht-Schmitt, T. E. Inorg. Chem. 2011, 50, 4842.

(8) (a) Tokunaga, T. K.; Kim, Y.; Wan, J. Environ. Sci. Technol. 2009 , 43, 5467. (b) Tokunaga, T. K.; Kim, Y.; Wan, J.; Yang, L. Environ. Sci. Technol. 2012 , 46, 7471.

(9) (a) Stern, T. W.; Stieff, L. R.; Girhard, M. N.; Meyrowitz, R. Am. Mineral. 1956 , 41, 187. (b) Baran, E. J.; Botto, I. L. Monatsh. Chem. 1976 , 107, 633. (c) Botto, I. L.; Baran, E. J. Z. Anorg. Allg. Chem. 1976 , 426, 321. (d) Frost, R. L.; Cejka, J.; Weier, M. L.; Martens, W.; Henry, D. A. Vib. Spectrosc. 2005, 39, 131.

(10) Chirayil, T.; Zavalij, P. Y.; Whittingham, M. S. Chem. Mater. 1998 , 10, 2629.

(11) (a) Dion, C.; Obbade, S.; Raekelboom, E.; Abraham, F.; Saadi, M. J. Solid State Chem. 2000 , 155, 342. (b) Obbade, S.; Dion, C.; Duvieubourg, L.; Saadi, M.; Abraham, F. J. Solid State Chem. 2003 , 173, 1. (c) Obbade, S.; Dion, C.; Rivenet, M.; Saadi, M.; Abraham, F. J. Solid State Chem. 2004 , 177, 2058. (d) Obbade, S.; Duvieubourg, L.; Dion, C.; Abraham, F. J. S*olid State Chem.* **200**7, 180, 866. (e) Obbade, S.; Renard, C.; Abraham, F. J. Solid State Chem. 2009 , 182, 413.

(12) (a) Jouffret, L.; Shao, Z. M.; Rivenet, M.; Abraham, F. J. Solid State Chem. 2010 , 183, 2290. (b) Rivenet, M.; Vigier, N.; Roussel, P.; Abraham, F. J. Solid State Chem. 2007 , 180, 713.

(13) Sheldrick, G. M. SHELXTL; Siemens Analytical X-ray Instruments: Madison, WI, 2001.

(14) Stoyanov, E. S.; Kim, K. C.; Reed, C. A. J. Am. Chem. Soc. 2006 , 128, 1948.

(15) Burns, P. C. Can. Mineral. 2005 , 43, 1839.

(16) Schindler, M.; Hawthorne, F. C.; Baur, W. H. Chem. Mater. 2000 , 12, 1248.

(17) Sullens, T. A.; Jensen, R. A.; Shvareva, T. Y.; Albrecht-Schmitt, T. E. J. Am. Chem. Soc. 2004 , 126, 2676.

(18) (a) Sarina, S.; Bo, A.; Liu, D.; Liu, H.; Yang, D.; Zhou, C.; Maes, N.; Komarneni, S.; Zhu, H. C*hem. Mater.* **2014**, 26, 4788. (b) Banerjee, C.; Dudwadkar, N.; Tripathi, S. C.; Gandhi, P. M.; Grover, V.; Kaushik, C. P.; Tyagi, A. K. J. Hazard. Mater. **2014**, 280, 63.

(19) Wang, S.; Diwu, J.; Simonetti, A.; Booth, C. H.; Albrecht-Schmitt, T. E. Environ. Sci. Technol. 2011, 45, 9457.

(20) (a) Binnemans, K.; Gorllerwalrand, C. Chem. Phys. Lett. 1995 , 235, 163. (b) Denning, R. G. J. Phys. Chem. A 2007 , 111, 4125.

(21) (a) Eaton, T.; Lin, J.; Cross, J. N.; Stritzinger, J. T.; Albrecht-Schmitt, T. E. Chem. Commun. 2014, 50, 3668. (b) Lin, J.; Diefenbach, K.; Fu, J.; Cross, J. N.; Clark, R. J.; Albrecht-Schmitt, T. E. Inorg. Chem. 2014 , 53, 9058.